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Journal of Molecular Catalysis A: Chemical 217 (2004) 203-210

www.elsevier.com/locate/molcata

Effect of annealing temperature on morphology, structure and photocatalytic behavior of nanotubed H₂Ti₂O₄(OH)₂

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Abstract

Nanotubed titanic acid ($H_2Ti_2O_4(OH)_2$) was prepared from nanotubed sodium titanate ($Na_2Ti_2O_4(OH)_2$) by an ion exchange reaction in a pH = 1 HCl solution. The effect of annealing temperature on the morphology, structure and photocatalytic behavior of nanotubed $H_2Ti_2O_4(OH)_2$ was studied by means of TEM, XRD, DTG, DSC, BET and ESR. The results showed that nanotubed $H_2Ti_2O_4(OH)_2$ is thermally unstable. Its dehydration consists of two steps. In the first-step dehydration, single-electron-trapped oxygen vacancies (SETOVs) were generated. Accompanying the second-step dehydration, the transition of crystal form from orthorhombic system to anatase took place, at the same time the nanotubes broke. At T > 300 °C, when the SETOV concentration greatly increased, the interaction between SETOV happened. (V_0°)_x formed could play the role of recombination center of photogenerated e^--h^+ and make the photocatalytic behavior of TiO₂ (anatase, obtained from 500 °C-treated nanotubed $H_2Ti_2O_4(OH)_2$) to become bad. © 2004 Elsevier B.V. All rights reserved.

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Keywords: Nanotubed titanic acid; TiO2; Photocatalytic activity; Single-electron-trapped oxygen vacancies

1. Introduction

Some alkaline metal titanates with different compositions and morphologies have been reported in literature, such as: tunnel structured Na₂Ti₁₆O₁₃ and zigzag structured Na₂Ti₃O₇ [1,2], layered Na₄Ti₉O₂₀·*x*H₂O [3], layered lithium titanate (Li_{1.81}H_{0.19})Ti₂O₅·2.2H₂O [4], fibrous K₂Ti₄O₉ [5], etc.

Titanic acids (such as $H_2Ti_3O_7$, $H_4Ti_9O_{20}$ · xH_2O , $H_2Ti_2O_5$ · $2.2H_2O$, $H_2Ti_4O_9$, etc.) could be obtained by treating the corresponding titanates with a 0.1–1.0 M HCl or HNO₃ [2–5]. At 700 °C, $H_4Ti_9O_{20}$ thermally decomposed to TiO₂ [3]:

 $\rm H_4Ti_9O_{20} \rightarrow 9TiO_2 + 2H_2O$

Layered (Li_{1.81}H_{0.19})Ti₂O₅·2.2H₂O showed a C-basecentered orthorhombic system with the lattice constants $a_0 = 16.66 \pm 0.02$ Å, $b_0 = 3.797 \pm 0.002$ Å and $c_0 =$ 3.007 ± 0.003 Å. Upon conversion to the hydrogen form the lattice constants changed to $a_0 = 18.08 \pm 0.03$ Å, $b_0 =$ 3.784 ± 0.003 Å and $c_0 = 2.998 \pm 0.002$ Å, respectively [4]. The photocatalytic properties of RuO₂-dispersed M₂Ti₆O₁₃ (M = Na, K, Rb, Cs) for water decomposition were investigated by Inoue and coworkers [6,7]. The water photosplitting activity depended on the kinds of alkaline metal atom and increased in the order of Na > K > Rb > Cs. Chatterjee et al. used density functional theory to rationalize the location and activity of ruthenium oxide incorporated into sodium hexatitanates [8].

A new nanotubed material was obtained by the reaction of polycrystalline TiO₂ with concentrated NaOH solution at 110 °C. From the contents of Na, Ti and structural water determined, it is concluded that the nanotubed material is Na₂Ti₂O₄(OH)₂ [9], rather than TiO₂, TiO_x or H₂TiO₃ [10–17]. After treatment with a HCl solution of pH = 1, nanotubed Na₂Ti₂O₄(OH)₂ can be converted to nanotubed H₂Ti₂O₄(OH)₂ [9]. The physico-chemical properties of both nanotubed Na₂Ti₂O₄(OH)₂ and H₂Ti₂O₄(OH)₂ are known little yet. Recently, we found that after appropriate treatment, nanotubed H₂Ti₂O₄(OH)₂ showed both strong visible light

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absorption and luminescence [18,19]. In this paper, the effect of annealing temperature on the morphology, structure and photocatalytic behavior of nanotubed $H_2Ti_2O_4(OH)_2$ is reported.

2. Experimental

2.1. Preparation of samples [9]

The 300 ml of 40% (w/w) NaOH aqueous solution was placed in a PTFE bottle, equipped with a reflux condenser. Then, the bottle was placed in an oil bath. When the NaOH solution was heated up to $110\,^\circ\text{C}$, 6 g anatase TiO₂ powder was added into it and stirred magnetically. After 20 h, the reaction stopped. When the dispersion cooled down to room temperature, it was diluted with de-ionized water to pH = 13.5. Part of the solid settled from the dispersion was washed with anhydrous ethanol for removing the free NaOH adsorbed on both outside and inside surface of the nanotubed material, then filtered and dried under vacuum at room temperature(sample nanotubed Na₂Ti₂O₄(OH)₂, Na/Ti atomic ratio = 1). The remained part was washed with de-ionized water to ca. neutral, filtered, then immersed in a pH = 1.0HCl solution for 5 h and washed with de-ionized water to remove Cl⁻, and dried under vacuum at room temperature (sample nanotubed $H_2Ti_2O_4(OH)_2$, Na/Ti atomic ratio = 0, H₂O/TiO₂ mole ratio close to 1). Heat treatment of samples was carried out in tubular furnace under air flow.

2.2. Characterization

Transmission electron microscopic (TEM) patterns were taken on a JEM-2010 electron microscope. X-ray diffraction (XRD) patterns were measured by a X'Pert Pro X-ray diffractometer. Differential thermogravimetric analysis (DTG) and differential scanning calorimetric measurement (DSC) were performed with a SEIKO Exstar-6000 thermal analysis system. BET specific surface areas were determined with a Micromeritics ASAP 2010 apparatus. Diffuse reflectance spectra (DRS) were recorded on a Shimadzu U-3010 spectrometer. Electron spin resonance (ESR) spectra were obtained on a Brüker ESP300E apparatus in air ambience. The *g*-tensors of ESR signals were obtained by taking g = 2.0036 of diphenyl picryl hydrazyl (DPPH) as reference.

2.3. Evaluation of photocatalytic behavior

The photocatalytic behavior of nanotubed $H_2Ti_2O_4(OH)_2$ was evaluated by propylene oxidation removal. The photocatalytic reactor was made of a flat quartz tube, and two 4 W black light lamps ($\lambda = 365 \text{ nm}$) were located outside the reactor. The catalyst was spread on both sides of surface-roughened glass plate. Feed gas of 500 ppm(v) propylene was made up of pure C_3H_6 and air, and was stored in a high pressure cylinder. The change of C_3H_6 concentration before and after reaction was determined by chromatographic method with hydrogen flame detector.

2.4. Reagents

Raw TiO₂ (anatase, specific surface area = $59 \text{ m}^2 \cdot \text{g}^{-1}$) was purchased from Hehai Nano-Technology Company, Jiangsu Province; NaOH, analytically pure, product of Tianjin Chemical Reagents Factory; HCl, analytically pure, product of Kaifeng Chemical Reagents Factory; anhydrous ethanol, analytically pure, produced by Tianjin Hongyan Reagents Factory.

3. Results and discussion

The morphologies of as-prepared nanotubed $Na_2Ti_2O_4$ (OH)₂ and $H_2Ti_2O_4(OH)_2$ are identical, which have a layered structure with the distance between the adjacent layers ca. 0.8 nm (Fig. 1a and b). Both show an orthorhombic



Fig. 1. TEM images of nanotube materials: (a) nanotubed $Na_2Ti_2O_4(OH)_2$ (insert is a four-layered nanotube with inner and outer diameters 6.4 and 9.3 nm, respectively); (b) nanotubed $H_2Ti_2O_4(OH)_2$ (insert is a multi-layered nanotube).

Table 1 XRD data of nanotubed Na₂Ti₂O₄(OH)₂

| 20 | d (Å) | hkl | <i>I</i> / <i>I</i> ₀ |
|-------|-------------------|-----|----------------------------------|
| 9.18 | 9.63 ^a | 200 | 100 |
| 24.30 | 3.66 | 110 | 32 |
| 28.14 | 3.17 | 600 | 60 |
| 34.24 | 2.62 (?) | 301 | 8 |
| 38.06 | 2.36 (?) | 501 | 10 |
| 48.14 | 1.89 | 020 | 69 |
| 61.76 | 1.50 (?) | 002 | 8 |

Lattice constants: $a_0 = 19.26 \text{ Å}$, $b_0 = 3.78 \text{ Å}$.

^a d_{200} is approximately equal to the distance between adjacent layers (0.8 nm) in Fig. 1a.

system with the lattice constants, $a_0 = 19.26$ Å, $b_0 = 3.78$ Å, but their peak intensities are different, especially for peaks (200), (110) and (600) (Table 1 and Fig. 2) [9,5,20], it may be a result of the replacement of Na⁺ ions in Na₂Ti₂O₄(OH)₂ by H⁺.

Figs. 3 and 4 indicate the TEM and XRD patterns of nanotubed Na₂Ti₂O₄(OH)₂ annealed at different temperatures for 2 h. It is obviously seen that at annealing temperature (*T*) \leq 500 °C its nanotube morphology and crystalline form remain unchanged, except the slight shift of d_{200} : i.e. 0.963 nm for as-prepared, 0.874 nm for 100–200 °C, 0.740 nm for 300–500 °C(see Fig. 4). The detailed discussion about the formation mechanism and structure of nanotubed Na₂Ti₂O₄(OH)₂ has been reported in our former paper [9]. The layered structure shown in Fig. 5a suggests that the decrease of d_{200} is related to the dehydration of interlayered OH groups:

$$Na_2Ti_2O_4(OH)_2 \rightarrow Na_2Ti_2O_5 + H_2O$$
(1)

Such a type of dehydration could reduce the interlayer distance but does not destroy the tubular shape. As for nanotubed H₂Ti₂O₄(OH)₂, at $T \leq 300$ °C there are a little change in nanotube length and (200) peak intensity (see



Fig. 2. XRD spectra of nanotube materials: (a) nanotubed $Na_2Ti_2O_4(OH)_2$; (b) nanotubed $H_2Ti_2O_4(OH)_2$; inset: hypothetical schematic view of orthorhombic system, *X*–*Y* direction.

Figs. 6a–d and 7a–d). But at T > 300 °C, accompanying with the formation of anatase phase [21], nanotubes break (Figs. 6e and f and 7e and f). These results reveal that the dehydration process of nanotubed H₂Ti₂O₄(OH)₂ is more complex than that of nanotubed Na₂Ti₂O₄(OH)₂. In Fig. 5b (Na⁺ replaced by H⁺), there are two types of dehydration: (i) dehydration of intralayered OH groups; (ii) dehydration



Fig. 3. TEM images of nanotubed $Na_2Ti_2O_4(OH)_2$ annealed in air at different temperatures for 2 h: (a) as-prepared (inset is a multi-layered nanotube); (b) 300 °C; (c) 500 °C (inset is a multi-layered nanotube).



Fig. 4. XRD spectra of nanotubed $Na_2Ti_2O_4(OH)_2$ annealed in air at different temperatures for 2 h: (a) as-prepared; (b) 100 °C; (c) 200 °C; (d) 300 °C; (e) 400 °C; (f) 500 °C.

of interlayered OH groups. At $T \leq 300$ °C the dehydration of intralayered OH groups caused a little change in nanotube length and (200) peak intensity. When annealing temperature >300 °C, the dehydration of interlayered OH groups induced the change of crystalline form from orthorhombic system to anatase, at the same time, nanotube morphology was destroyed. These two types of dehydration reactions can be formulated as follows:

$$H_2 Ti_2 O_4 (OH)_2 \rightarrow H_2 O + H_2 Ti_2 O_5$$
⁽²⁾

$$H_2 Ti_2 O_5 \rightarrow H_2 O + 2 Ti O_2 (anatase)$$
(3)



Fig. 5. Schematic diagram of layered composition (X-Y section): (a) nanotubed Na₂Ti₂O₄(OH)₂; (b) nanotubed H₂Ti₂O₄(OH)₂.

M. Zhang et al. / Journal of Molecular Catalysis A: Chemical 217 (2004) 203-210



Fig. 6. TEM images of nanotubed $H_2Ti_2O_4(OH)_2$ annealed in air at different temperatures for 2 h: (a) as-prepared (inset is a multi-layered nanotube); (b) 100 °C; (c) 200 °C; (d) 300 °C; (e) 400 °C; (f) 500 °C.



Fig. 7. XRD spectra of nanotubed $H_2Ti_2O_4(OH)_2$ annealed in air at different temperature for 2 h: (a) as-prepared; (b) 100 °C; (c) 200 °C; (d) 300 °C; (e) 400 °C; (f) 500 °C.



Fig. 8. Thermochemical property of nanotubed H₂Ti₂O₄(OH)₂: (a) TG and DTG curves; (b) DSC curve.

DTG, DSC and specific surface area (S) data provide further evidences for above two types of dehydration. In Fig. 8a, there are two broad DTG peaks: broad peak I (room temperature to 300 °C, peak at 79.8 °C) consists of the desorption of physically adsorbed water and intralayered water (formula (2)), this peak is corresponding to the endothermic peak on DSC curve (Fig. 8b); broad peak II (300-500 °C, peak at 390.9 °C) represents the desorption of interlayered water (formula (3)), this peak is corresponding to the exothermic peak on DSC curve (Fig. 8b) which reveals that the transition of crystal form from orthorhombic system to anatase is exothermic. Fig. 9 indicates the dependence of specific surface area of nanotubed H2Ti2O4(OH)2 on annealing temperature. In the range of room temperature to 300 °C, S decreases slowly $(S_{R,T} = 402 \text{ m}^2 \text{ g}^{-1}, S_{300} =$ $354 \text{ m}^2 \text{ g}^{-1}$; at $T > 300 \,^{\circ}\text{C}$, the nanotubes break, then S decreases sharply $(S_{400} = 190 \text{ m}^2 \text{ g}^{-1}, S_{500} = 118 \text{ m}^2 \text{ g}^{-1}, S_{600} = 60 \text{ m}^2 \text{ g}^{-1}).$

DRS spectra (Fig. 10, inset) of both as-prepared nanotubed H₂Ti₂O₄(OH)₂ and 500 °C-treated nanotubed H₂Ti₂O₄(OH)₂ are similar to raw TiO₂, however, their photocatalytic behavior are greatly different. The results shown in Figs. 6–8 have proved that: $T \le 300$ °C, H₂Ti₂O₅ forms (formula (2)), T > 300 °C, the transition of crystal form takes place (formula (3)). Fig. 10a shows that the photocatalytic activity of both H₂Ti₂O₄(OH)₂ and H₂Ti₂O₅ are very low, once the formation of anatase phase begins, the photocatalytic activity increases dramatically and reaches a maximum at ca. T = 500 °C (the dependence of photocatalytic activity of nanotubed Na₂Ti₂O₄(OH)₂ on annealing temperature was also determined, both Na₂Ti₂O₄(OH)₂ and Na₂Ti₂O₅ are inert (Fig. 10b) for this reaction). It should be



Fig. 9. Specific surface area (S) of nanotubed $H_2Ti_2O_4(OH)_2$ annealed at different temperatures for 2 h (as-prepared sample evacuated at room temperature for 20 h).

pointed out that the photocatalytic activity of $500 \,^{\circ}$ C-treated nanotubed H₂Ti₂O₄(OH)₂ (i.e. TiO₂, anatase) is only about 60% of that of raw TiO₂ (point c in Fig. 10). For explaining this phenomenon, ESR signals of nanotubed H₂Ti₂O₄(OH)₂ treated at different temperatures were measured in air ambi-

ence (see Fig. 11). The results show that along with the increase of annealing temperature, the ESR signal varies from a single symmetrical peak (g = 2.003) to an unsymmetrical peak ($g_{\perp} = 2.005$, $g_{\parallel} = 1.985$), the transition temperature of ESR behavior coincides with that of the crystalline form and morphology shown in Figs. 7 and 6, respectively. From Fig. 5b we can see that the intralayered dehydration of nanotubed H₂Ti₂O₄(OH)₂ produce two types of lattice defects: oxygen vacancy and hydrogen vacancy. According to the principle of neutrality and Kröger and Vink's symbols, it may be expressed as follows [26]:

$$2OH \rightarrow H_2O + V_O^{o} + V'_H \tag{4}$$

where V_0^{o} is the single-electron-trapped oxygen vacancy (SETOV) with one effective positive charge, V'_H is the hydrogen vacancy with one effective negative charge.

It was known that oxygen vacancy trapped one electron can be described by a model of a singly-ionized helium-like atom and shows a symmetrical ESR signal with g = 2.003[18,22–25]. Fig. 11b–d evidence the existence of SETOV. At T > 300 °C, when the SETOV concentration greatly increased, the interaction between SETOV happened [26]:

$$x \mathbf{V}_{\mathbf{O}}^{\mathbf{O}} \rightleftharpoons (\mathbf{V}_{\mathbf{O}}^{\mathbf{O}})_{x} \tag{5}$$

 $(V_0^{\circ})_x$ formed shows an unsymmetrical ESR signal with $g_{\perp} = 2.005$, $g_{\parallel} = 1.985$ (see Fig. 11e–g), which could play



Fig. 10. Dependence of photocatalytic behavior on annealing temperature: (a) nanotubed $H_2Ti_2O_4(OH)_2$; (b) nanotubed $Na_2Ti_2O_4(OH)_2$; (c) raw TiO_2 (inset: DRS spectra).



Fig. 11. ESR spectra of nanotubed H₂Ti₂O₄(OH)₂ annealed at different temperatures for 2 h: (a) as-prepared; (b) 100 °C; (c) 200 °C; (d) 300 °C; (e) 400 °C; (f) 500 °C; (g) 600 °C (g = 2.003, $g_{\perp} = 2.005$, $g_{\parallel} = 1.985$).

the role of recombination center of photogenerated e^{-} -h⁺ and make the photocatalytic behavior of TiO₂ (anatase, obtained from 500 °C-treated nanotubed H₂Ti₂O₄(OH)₂) to become bad (compared with raw TiO₂). As for the decrease of activity at 550 °C (Fig. 10a), it should be resulted from the further decrease of S after the formation of anatase phase (see Fig. 9).

4. Conclusions

The morphology and structure of nanotubed titanic acid $(H_2Ti_2O_4(OH)_2)$ is thermally unstable. Its dehydration consists of two steps: (i) dehydration of intralayered OH groups at annealing temperature $\leq 300 \,^{\circ}$ C; (ii) dehydration of interlayered OH groups at annealing temperature $>300 \,^{\circ}$ C. In the first-step dehydration, V_0° are generated. Accompanied with the second-step dehydration, the transition of crystal form from orthorhombic system to anatase takes place, at the same time, nanotubes break and the interaction between SETOV happens. $(V_0^{\circ})_x$ formed could play the role of recombination center of photogenerated e^--h^+ , and make the photocatalytic behavior of TiO₂ (anatase, obtained from 500 $^{\circ}$ C-treated nanotubed H₂Ti₂O₄(OH)₂) to become bad.

Acknowledgements

We acknowledge the support of the National Natural Science Foundation of China (No. 20071010). The authors are indebted to Porf. Xu Yuanzhi and Chen Jingrong's help for the measurements and discussion of ESR data.

References

- [1] S. Andersson, Acta Crystallogr. 15 (1962) 194-201.
- [2] H. Izawa, S. Kikkawa, M. Koizymi, J. Phys. Chem. 86 (1982) 5023– 5026.
- [3] A. Clearfield, J. Lehto, J. Solid State Chem. 73 (1988) 98–106.
- [4] M. Sugita, M. Tsuji, M. Abe, Bull. Chem. Soc. Jpn. 63 (1990) 1978–1984.
- [5] T. Sasaki, M. Watanabe, Y. Komatsu, Y. Fujiki, Inorg. Chem. 24 (1985) 2265–2271.
- [6] S. Ogura, M. Kohno, K. Sato, Y. Inoue, Phys. Chem. Chem. Phys. 1 (1999) 179–183.
- [7] Y. Inoue, T. Kubokawa, K. Sato, J. Phys. Chem. 95 (1991) 4059– 4063.
- [8] A. Chatterjee, H. Hayashi, T. Iwasaki, J. Phys. Chem. B 105 (2001) 3463–3469.
- [9] J. Yang, Z. Jin, X. Wang, W. Li, J. Zhang, S. Zhang, X. Gao, Z. Zhang, Dalton Trans. (2003) 3898–3901.
- [10] T. Kasuga, M. Hiramatsu, A. Hoson, T. Sekino, K. Niihara, Langmuir 14 (1998) 3160–3163.
- [11] S. Zhang, J. Zhou, Z. Zhang, A.V. Vorontsov, Z. Jin, Chin. Sci. Bull. 45 (16) (2000) 1533–1536.
- [12] G.H. Du, Q. Chen, R.C. Che, Z.Y. Yuan, L.M. Peng, Appl. Phys. Lett. 79 (2001) 3702–3704.
- [13] B.D. Yao, Y.F. Chan, X.Y. Zhang, W.F. Zhang, Z.Y. Yang, N. Wang, Appl. Phys. Lett. 82 (2003) 281–283.
- [14] Q. Chen, G.H. Du, L.M. Peng, J. Chin. Electron. Microsc. Soc. 21 (2002) 265–269.
- [15] T. Kasuga, M. Hiramatsu, A. Hoson, T. Sekino, K. Niihara, Adv. Mater. 11 (1999) 1307–1311.
- [16] Y. Zhu, H. Li, Y. Koltylpin, Y.R. Hacohen, A. Gedanken, Chem. Commun. (2001) 2616–2617.
- [17] J. Zhang, X. Guo, Z. Jin, S. Zhang, J. Zhou, Z. Zhang, Chin. Chem. Lett. 14 (2003) 419.
- [18] L. Qian, Z.S. Jin, J.W. Zhang, Y.B. Huang, Z.J. Zhang, Z.L. Du, Appl. Phys. A, published online: 11 February 2001.
- [19] S.L. Zhang, W. Li, Z.S. Jin, J.J. Yang, J.W. Zhang, Z.L. Du, Z.J. Zhang, J. Solid State Chem. 177 (2004) 1365–1371.
- [20] Joint Committee on Powder Diffraction Standard, 47-0124.
- [21] Joint Committee on Powder Diffraction Standard, 84-1286.
- [22] E. Serwicka, M.W. Schlierkamp, R.N. Schindler, Z. Naturforch. 36a (1981) 226–232.
- [23] E. Serwicka, Colloids Surf. 13 (1985) 287-293.
- [24] D.C. Cronemeyer, Phys. Rev. 113 (1959) 1222-1226.
- [25] I. Nakamura, N. Negishi, S. Kutsuna, T. Ihara, S. Sugihara, K. Takeuchi, J. Mol. Catal. A: Chem. 161 (2000) 205–212.
- [26] M. Su, Solid State Chemistry: An Introduction, Peking University Press, Beijing, 1987 (in Chinese).